

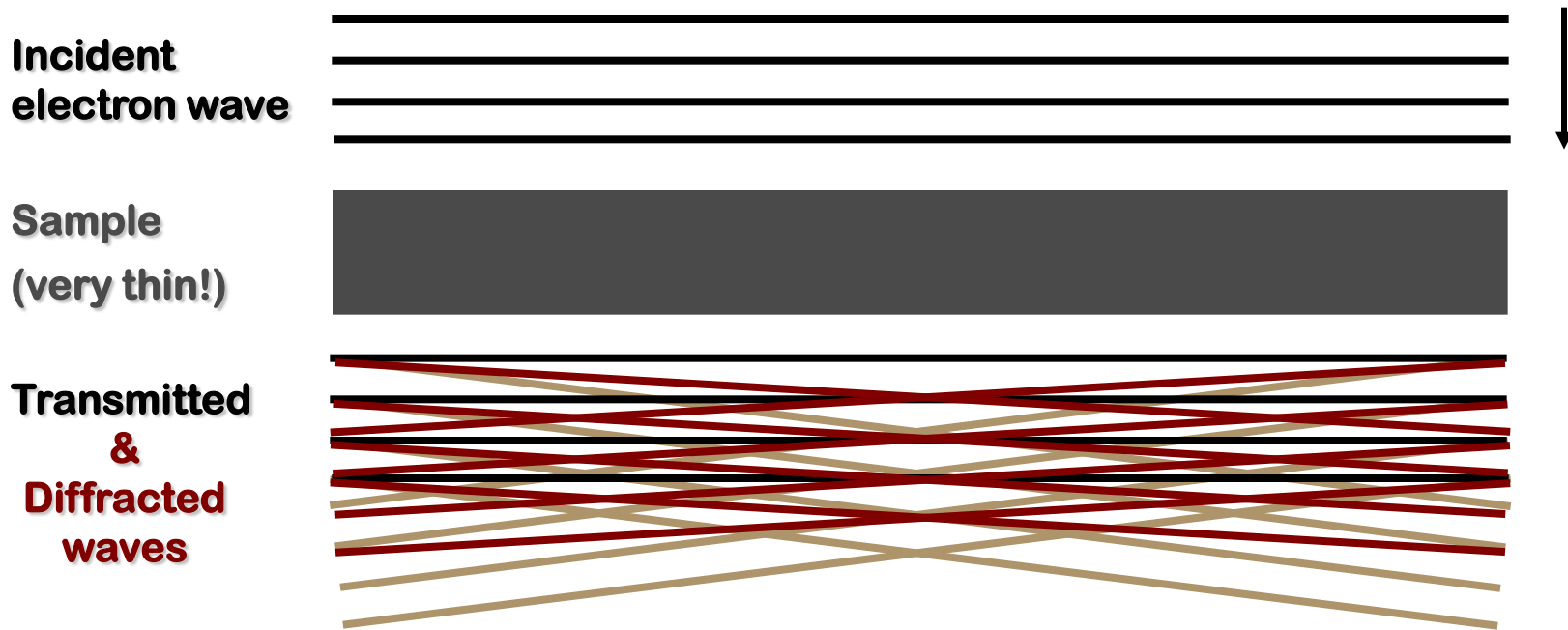
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# **Overview of High-Resolution TEM & Scanning TEM**

## **Lecture 11**

# High-resolution EM

## *general idea*



**Transmitted & diffracted waves each have a different phase**

**Result is an interference pattern - our 'phase contrast' or HREM image**

# High-resolution EM

*general idea*

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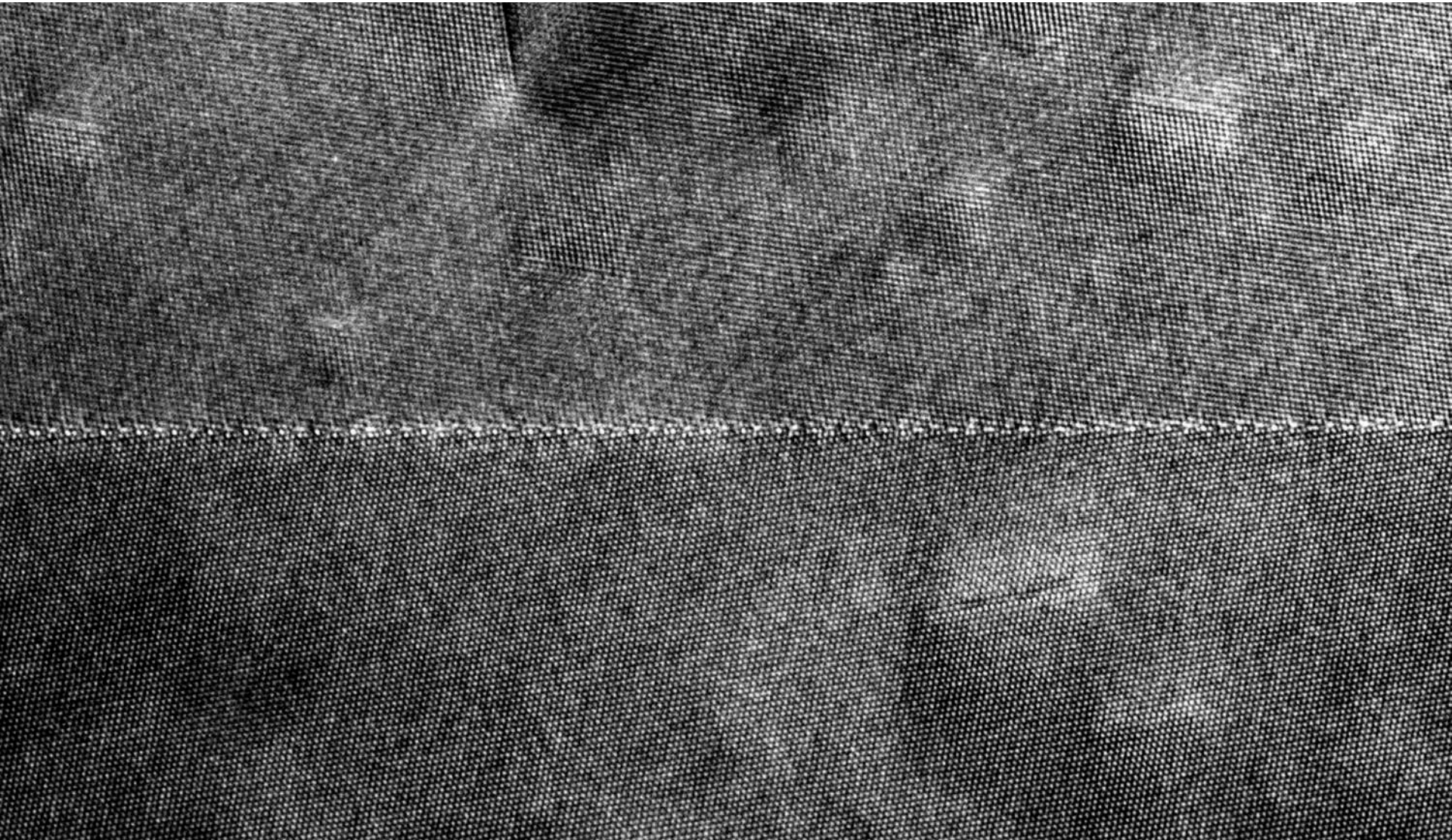


Image courtesy U. Dahmen, NCEM, LBNL

# High-resolution EM

*general idea*

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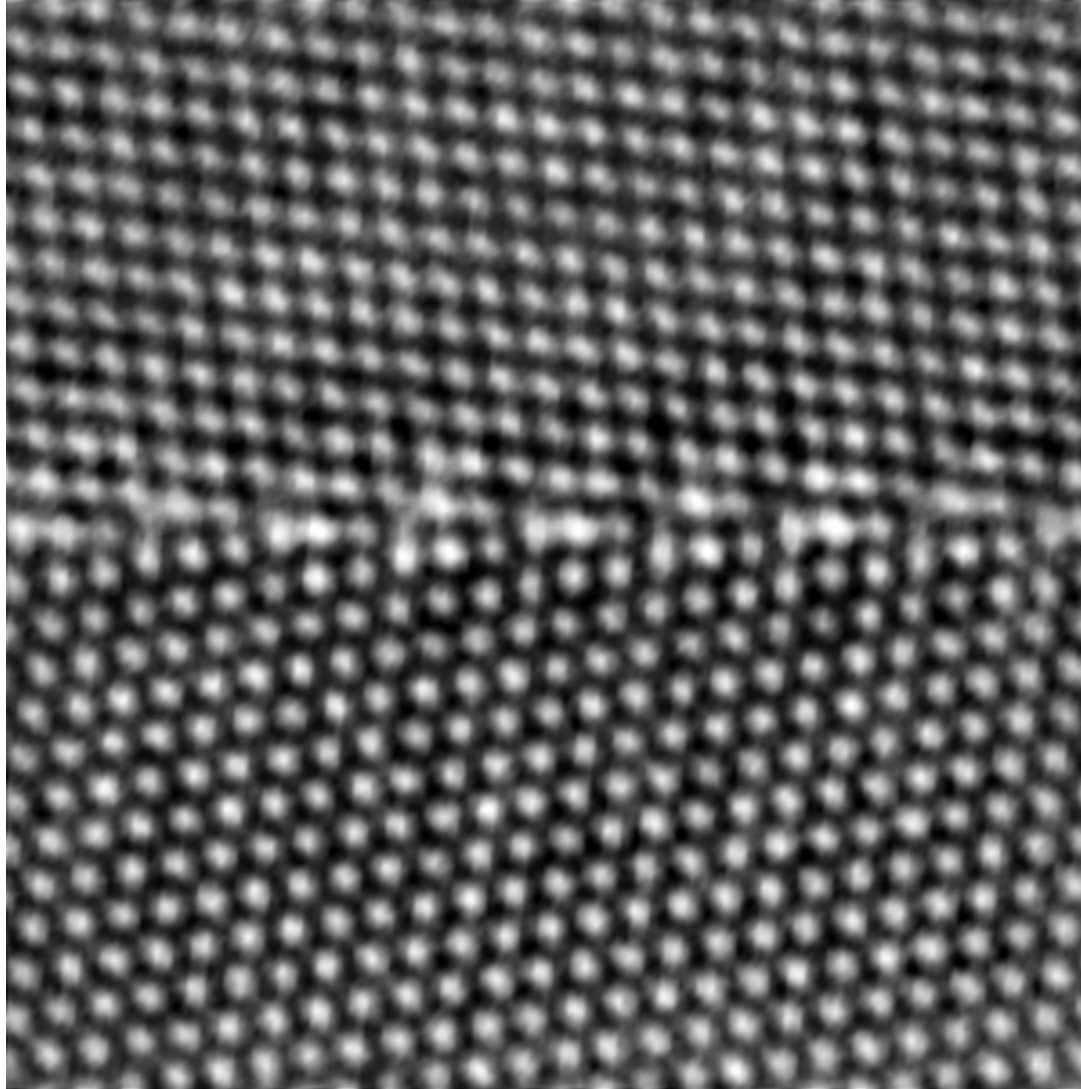


Image courtesy U. Dahmen, NCEM, LBNL

# High-resolution EM

## *general idea*

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**Why are the phases different?**

**Transmitted & diffracted waves travel different distances in the crystal**

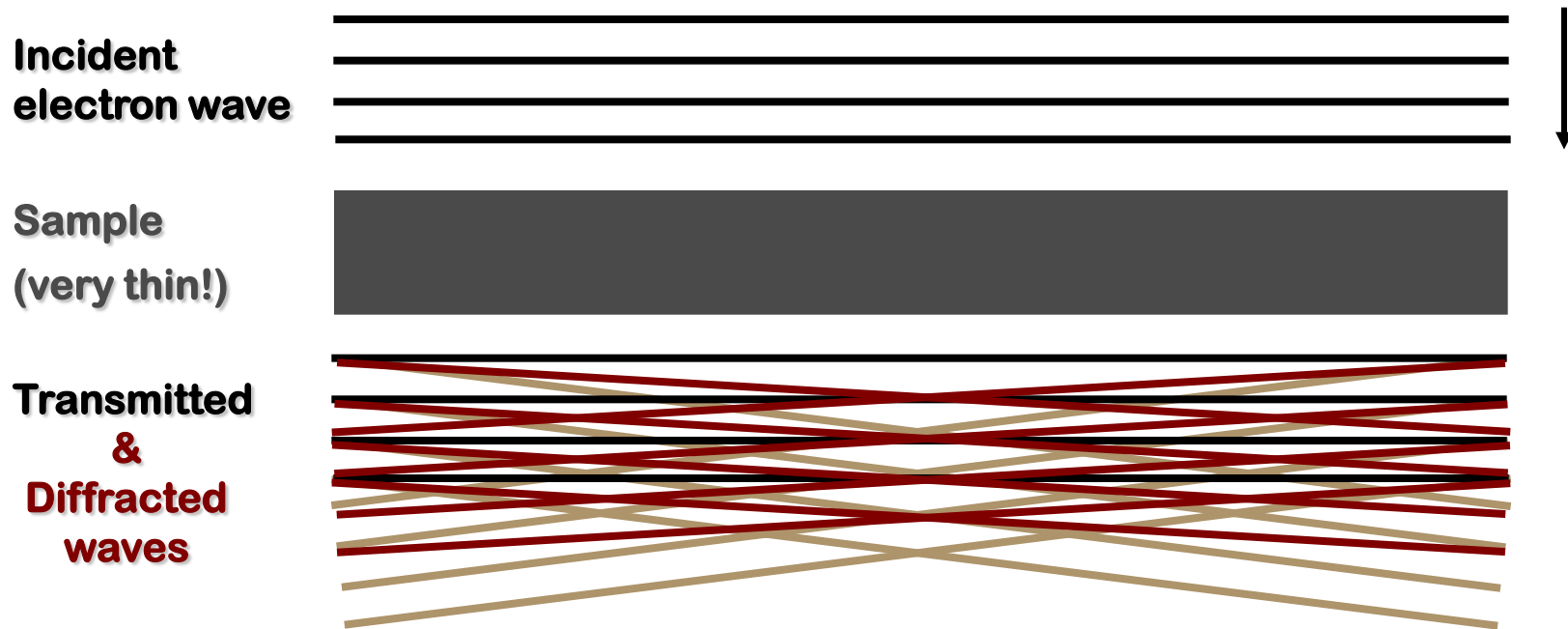
**Each diffracted wave will have its own phase**

**Highly simplified explanation: there is much more to this, but this conveys the idea**

- In fact, it is because each diffracted wave represents a different solution to the Schrödinger Eqn. for the electron in the crystal**
  - Resulting phase depends on the strength & spacing of the periodic potential of the lattice along a given direction in the crystal**

# High-resolution EM

## *general idea*



So, appears “simple” enough ...

- (1) Calculate the phase differences for the different diffracted waves (not easy, but do-able)
- (2) Create an interference pattern from the overlap of these phases in two-dimensions

# High-resolution EM

## *general idea*

Not even this “simple”

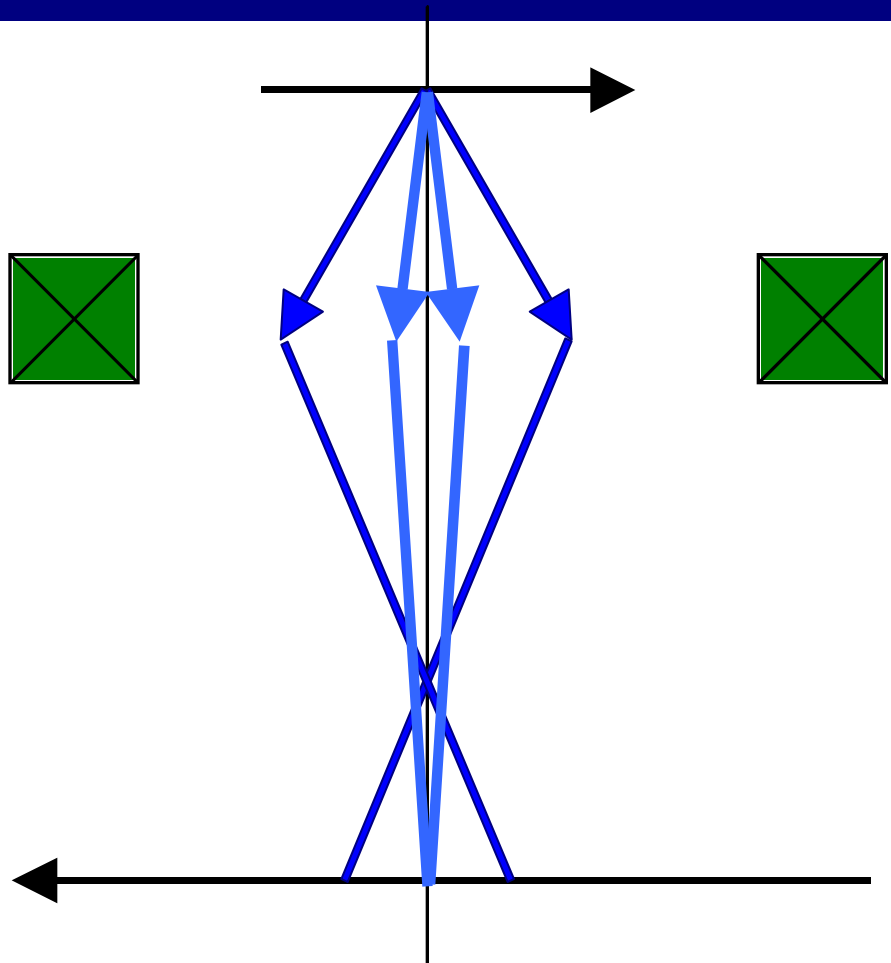
The TEM has very poor lenses

- Spherical aberration in particular

This aberration causes diffracted waves to be ‘phase shifted’ by the objective lens

- Complex dependence on wavelength,  $C_s$ , diffraction vector and defocus
- Magnitude of phase shift varies with distance from optic axis
  - And thus diffraction angle
  - Thus each diffracted wave undergoes a different phase shift

Complicates image interpretation

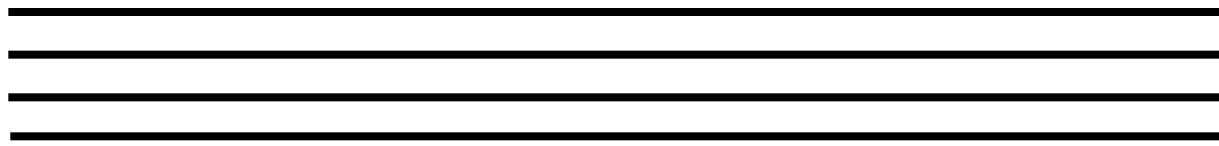


Spherical aberration

# High-resolution EM

*general idea*

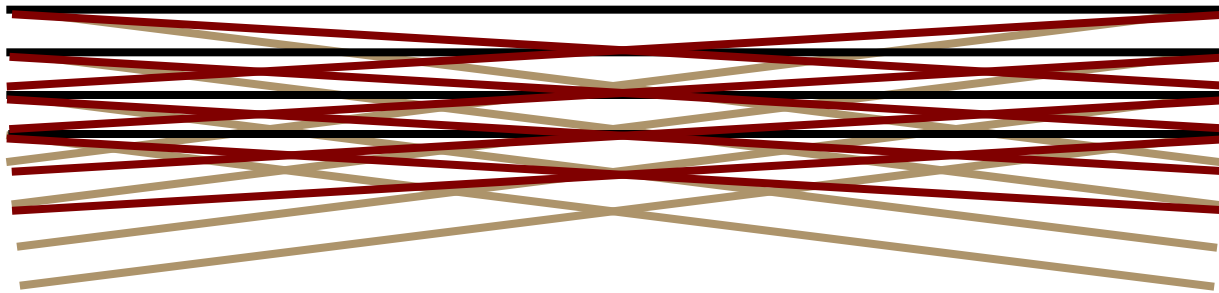
Incident  
electron wave



Sample  
(very thin!)



Transmitted  
&  
Diffracted  
waves



Returning to this picture

This means that the phases of the diffracted waves are  
changed by the objective lens focus

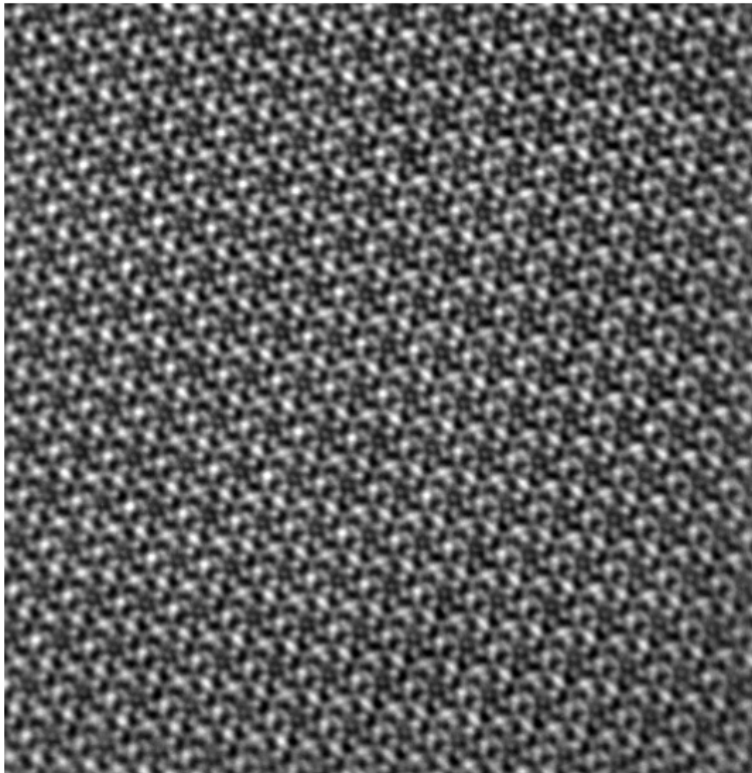


# High-resolution EM

## *general idea*

Thus, the image you get **STRONGLY DEPENDS ON THE FOCUS CONDITION**

A single HREM image



The 'unscrambled' exit wave

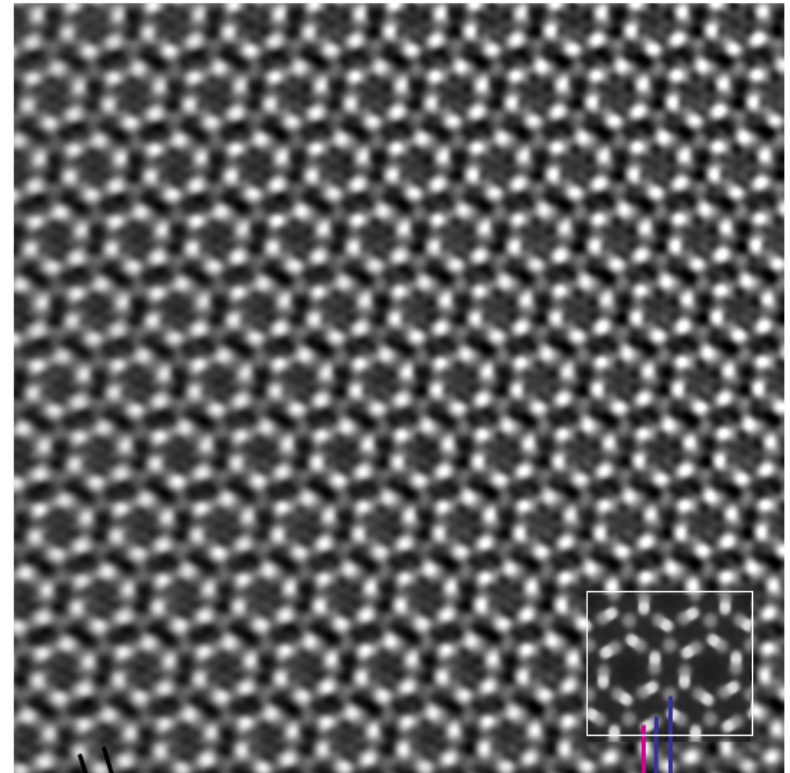


Image courtesy C. Kisielowski, NCEM, LBNL

0.18 nm

Simulation: Si N<sub>1,2</sub>

# High-resolution EM

## *general idea*

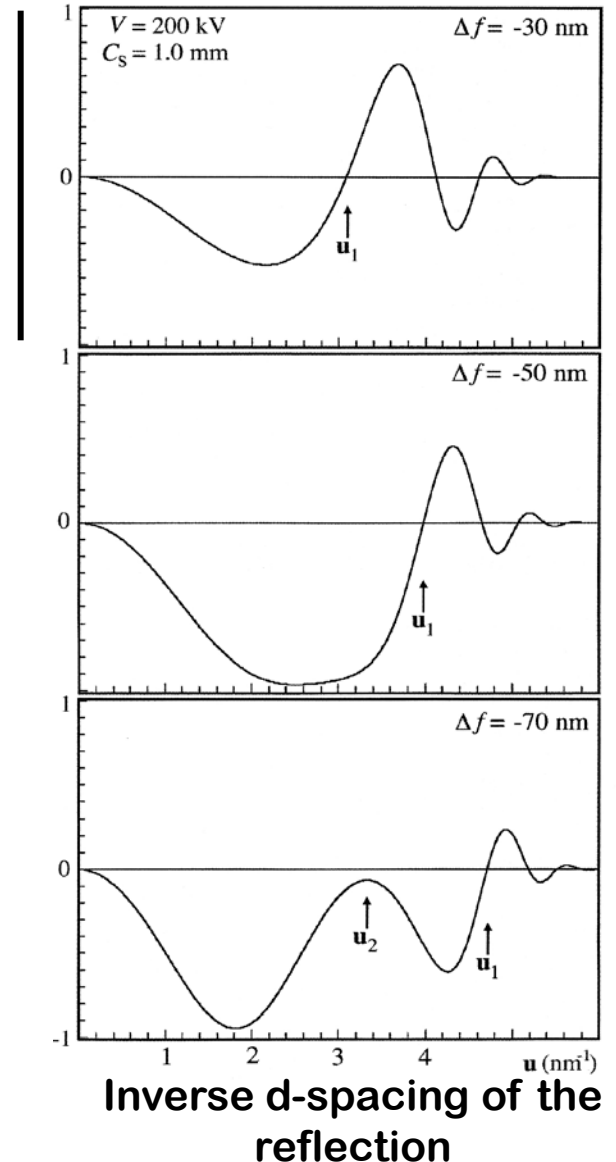
So, the lens effectively ‘scrambles’ the information embedded in the exit wave

The amount of scramble depends on the defocus &  $C_s$

Embodied in the ‘Contrast Transfer Function’

Different diffracted waves undergo different modifications of their spatial frequencies

“relative information transfer”



# Scherzer defocus

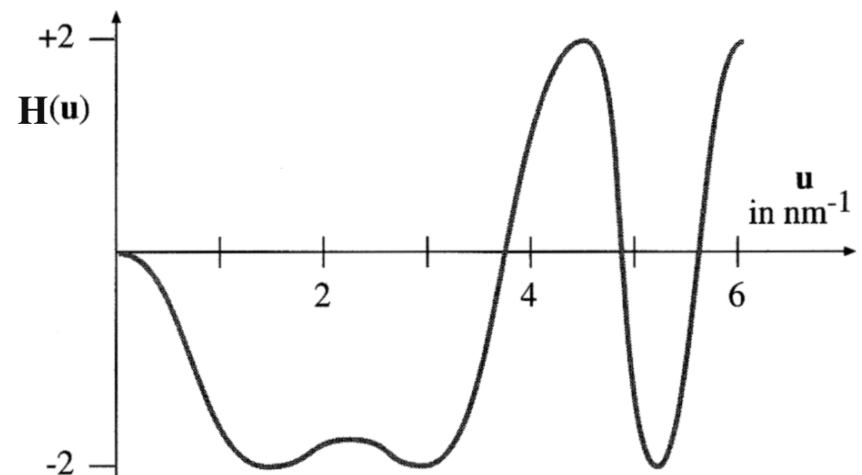
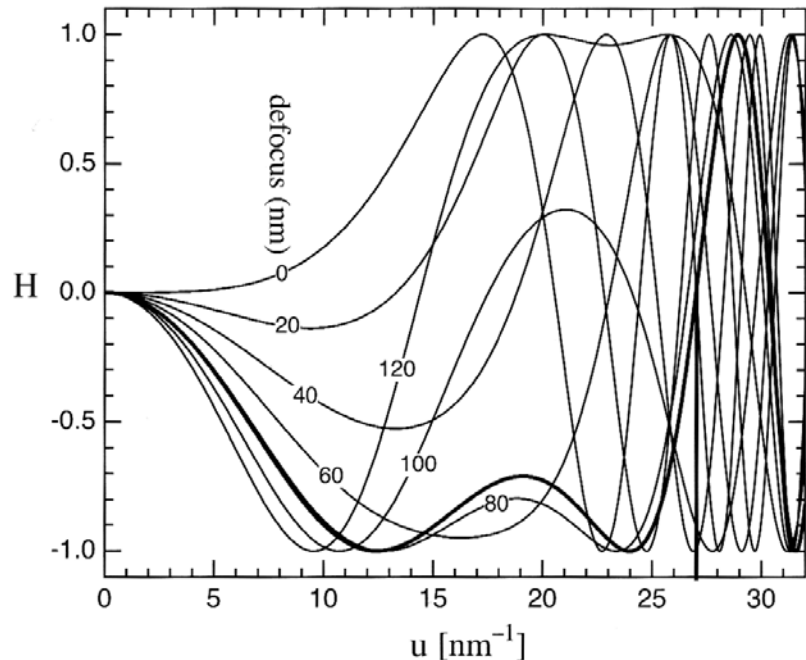
**Balance effect of spherical aberration with a specific value of negative defocus.**

**Scherzer defocus:**

$$\Delta f = -\left(\frac{4}{3} C_s \lambda\right)^{1/2}$$

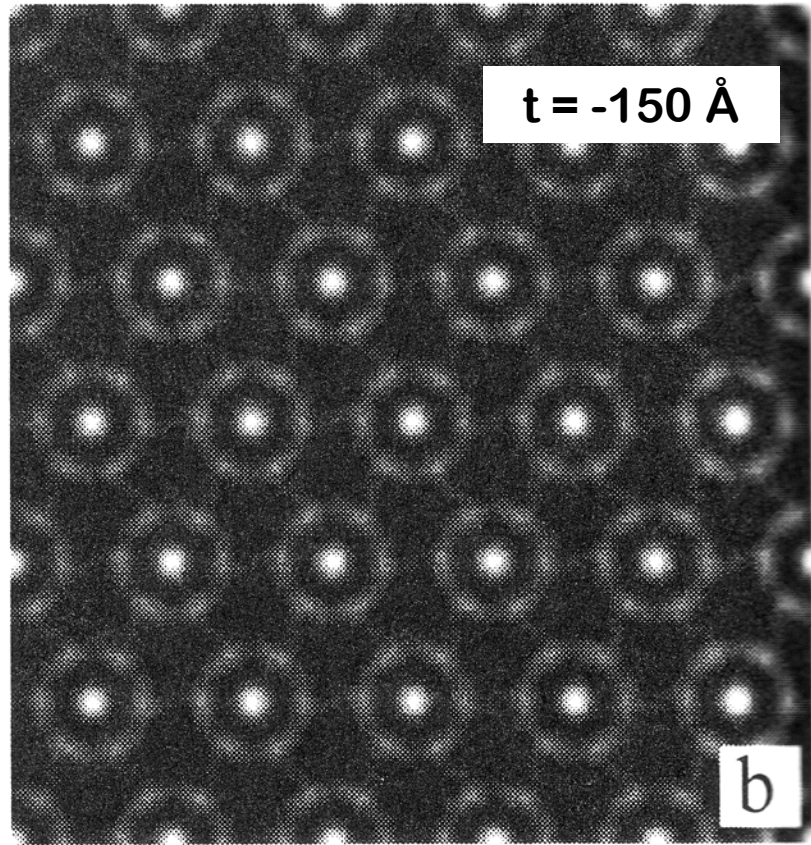
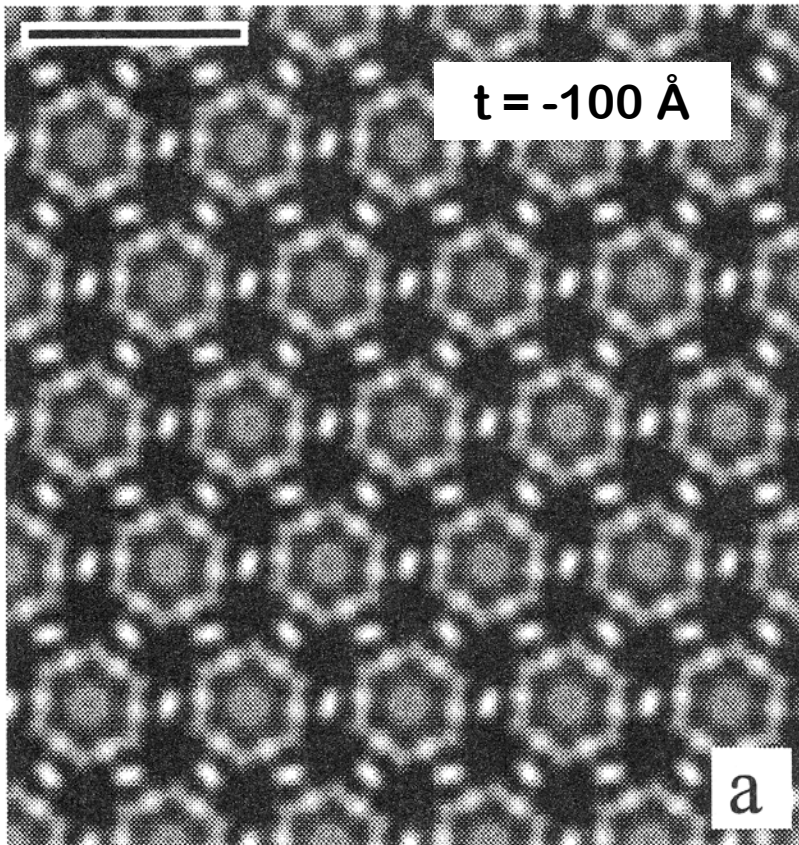
**Scherzer resolution:**

$$R_{\text{Scherzer}} = \frac{1}{1.51} C_s^{-1/4} \lambda^{-3/4}$$



# Example

$Si_3N_4$  (0001)



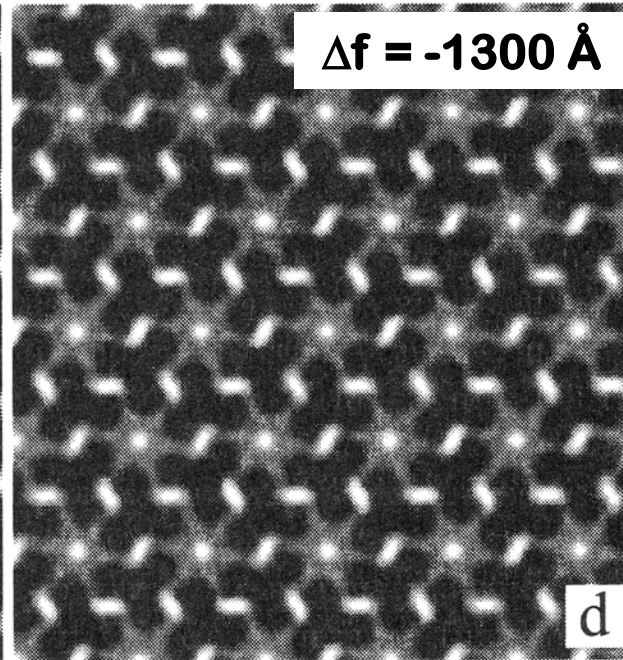
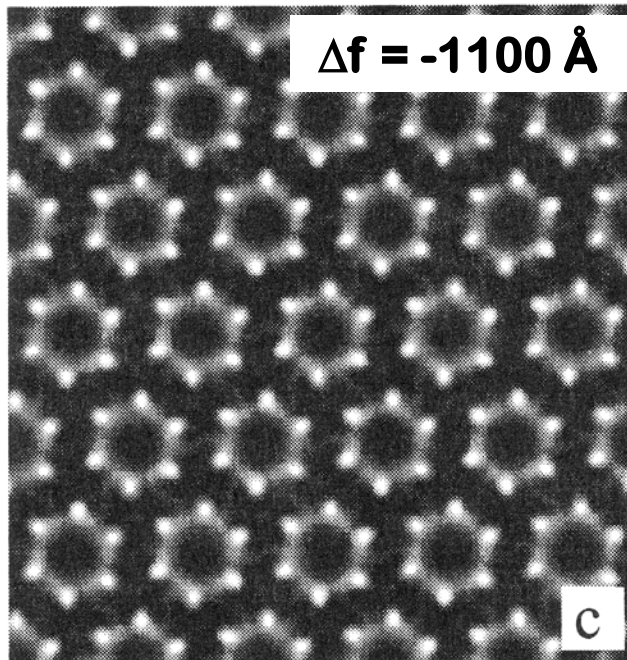
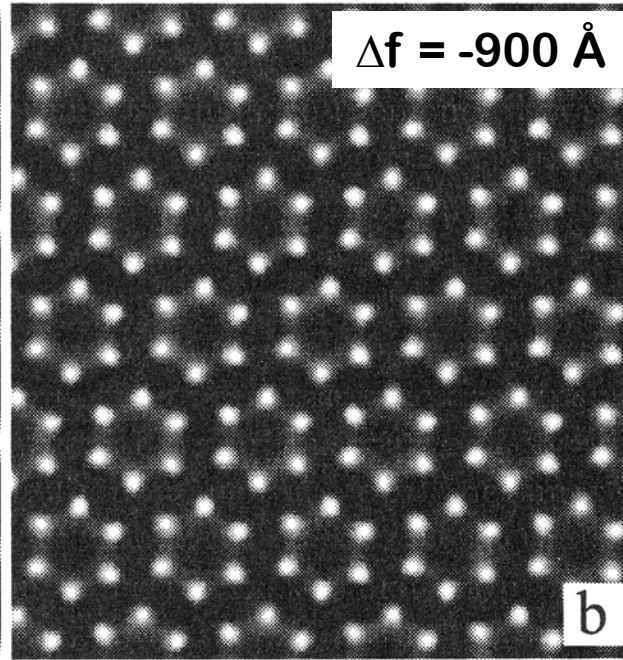
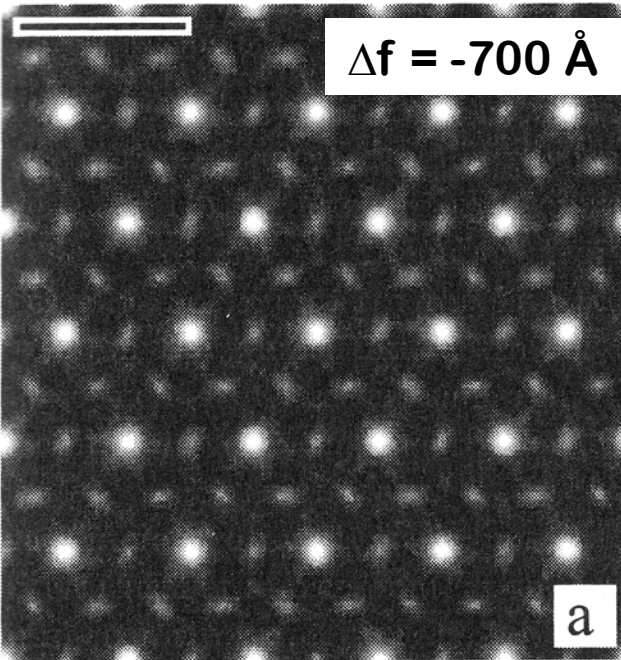
**Images depend on sample thickness  
(different phase shift as electron wave travels a different distance)**

# Example

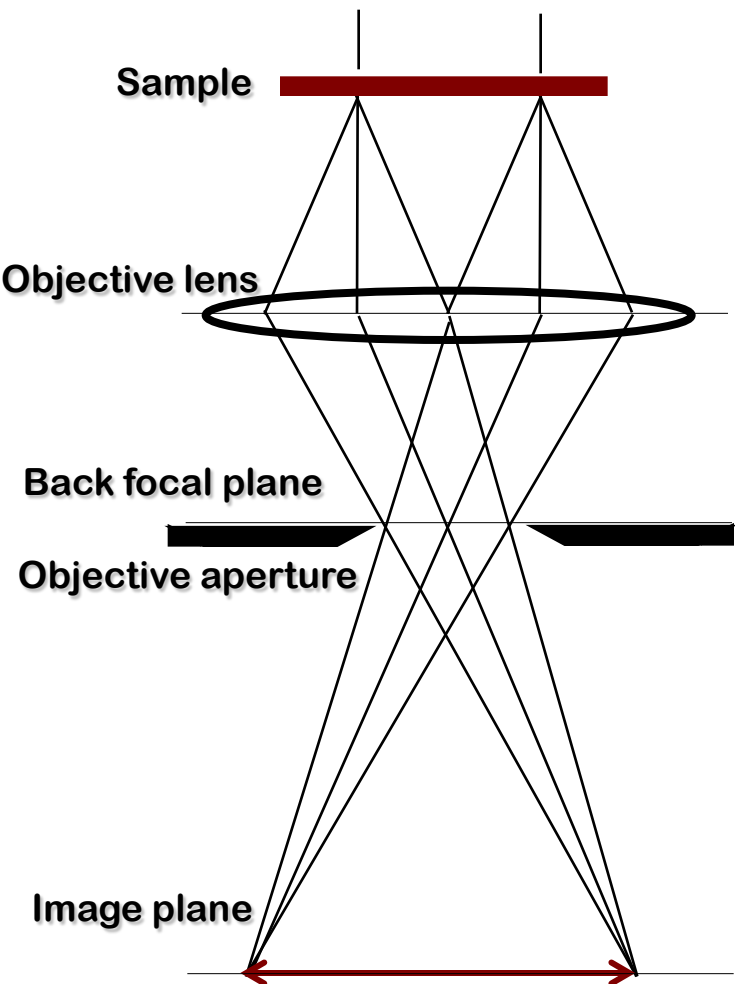
*Si<sub>3</sub>N<sub>4</sub> (0001)*

Images depend on focus

Different relative phase shifts of the diffracted waves with respect to the transmitted wave



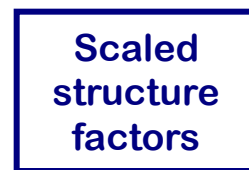
# Plane Ray Diagram



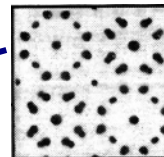
## Image simulation

# Function

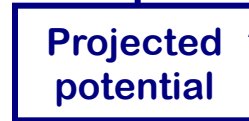
# Display



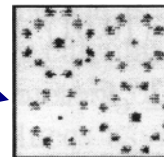
$V(x,y)$



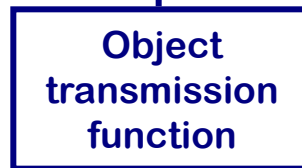
Model structure



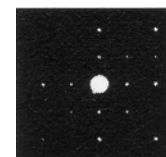
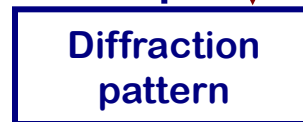
$\phi(x,y)$



Projected potential



Multislice calculation



Diffraction Intensity

Contrast Transfer Function



$\Psi(x,y)$

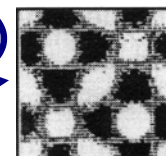


Image Intensity

# Summary - HRTEM

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**Image strongly depends on defocus**

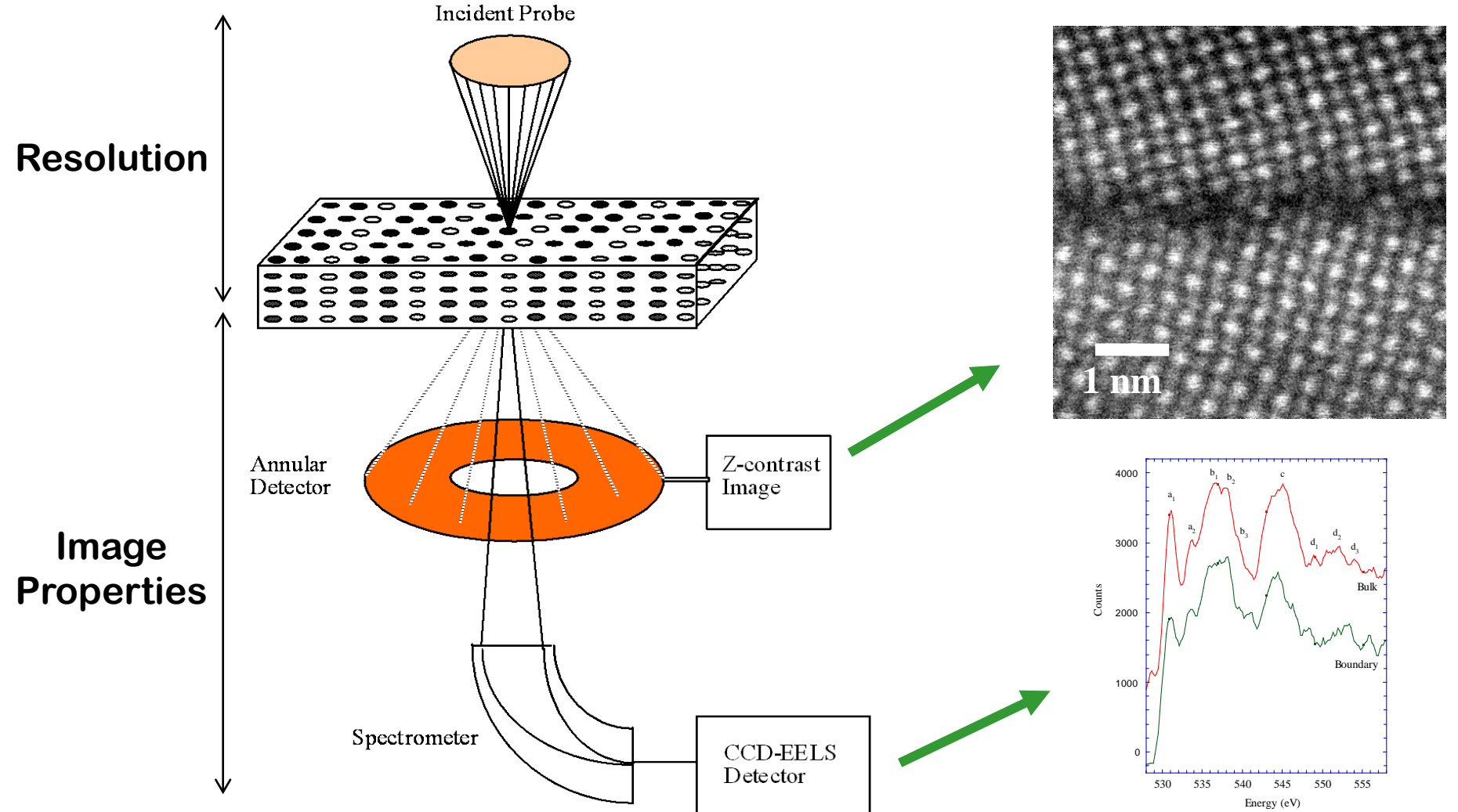
**Relationship between image and “atomic positions” is not straightforward**

**Understanding of imaging conditions (via defocus, sample thickness and microscope calibrations) necessary**

**These provide inputs for image simulation**

**Proper match of the image with the calculation required for true understanding of the image**

# STEM general idea





# Types of STEM images

## Bright-field

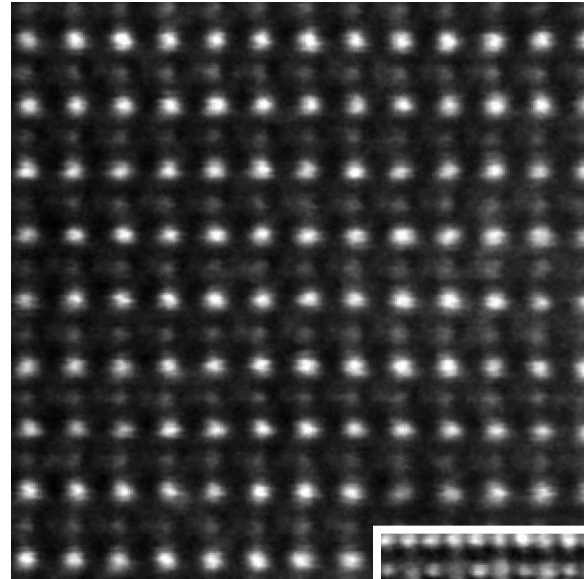
- Collect central beam with a small collection angle
- Contains elastic (Rutherford), phonon, plasmon and Compton

## Low-angle annular dark field

- Collection angle of 25 - 50 milliradians (mrad)
- Mostly phonon scatter

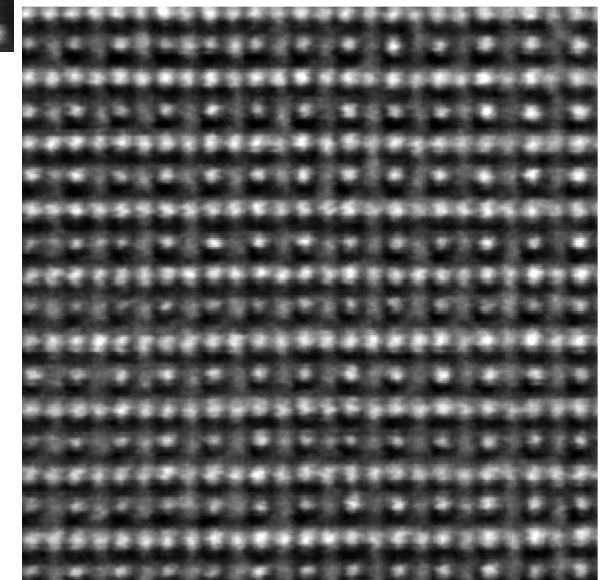
## High-angle annular dark field

- Collection angle of 50 - 250 mrad
- Largely phonon scatter (TDS)

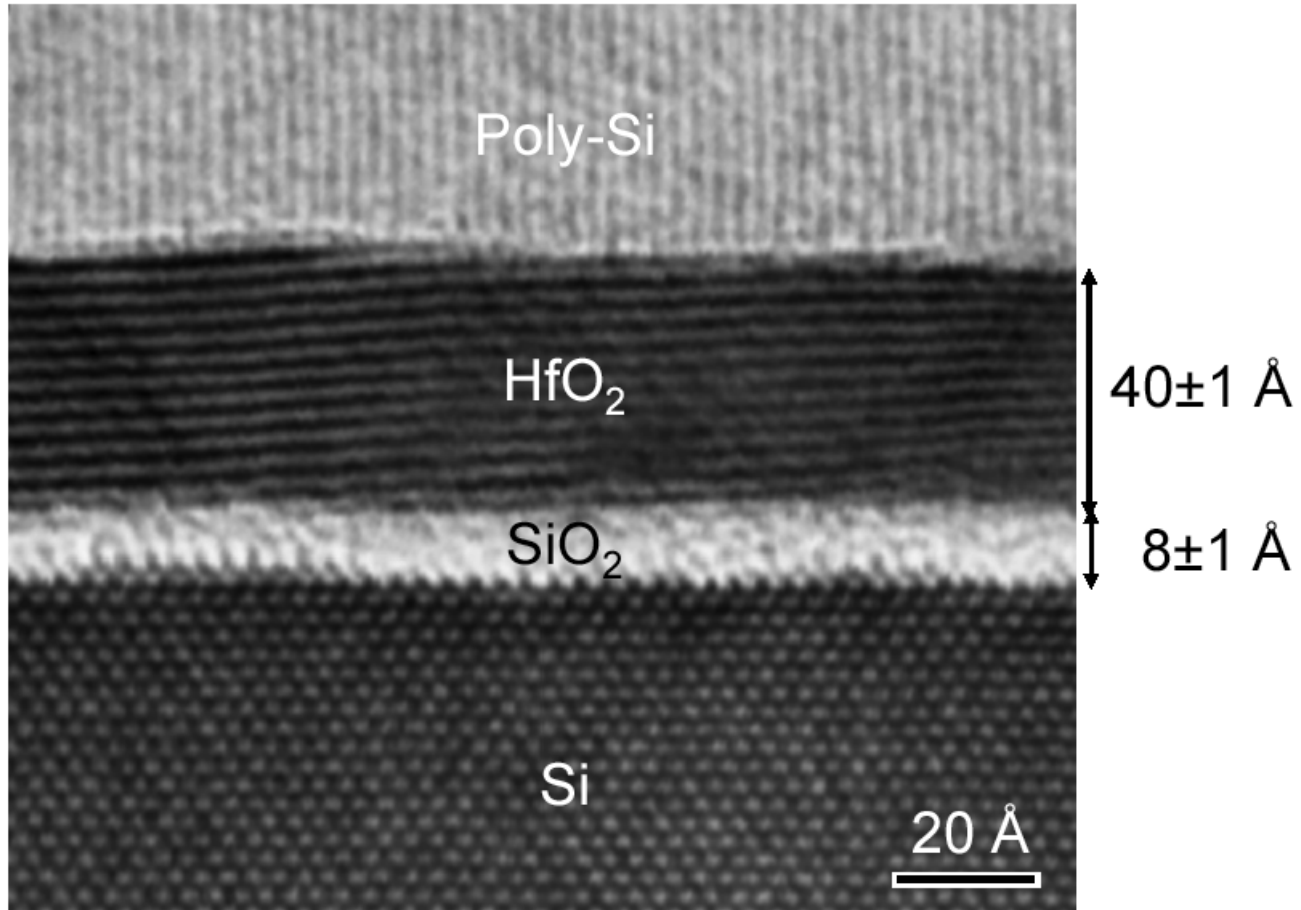


Coherent BF-STEM image of SrTiO<sub>3</sub> <110>

HAADF-STEM

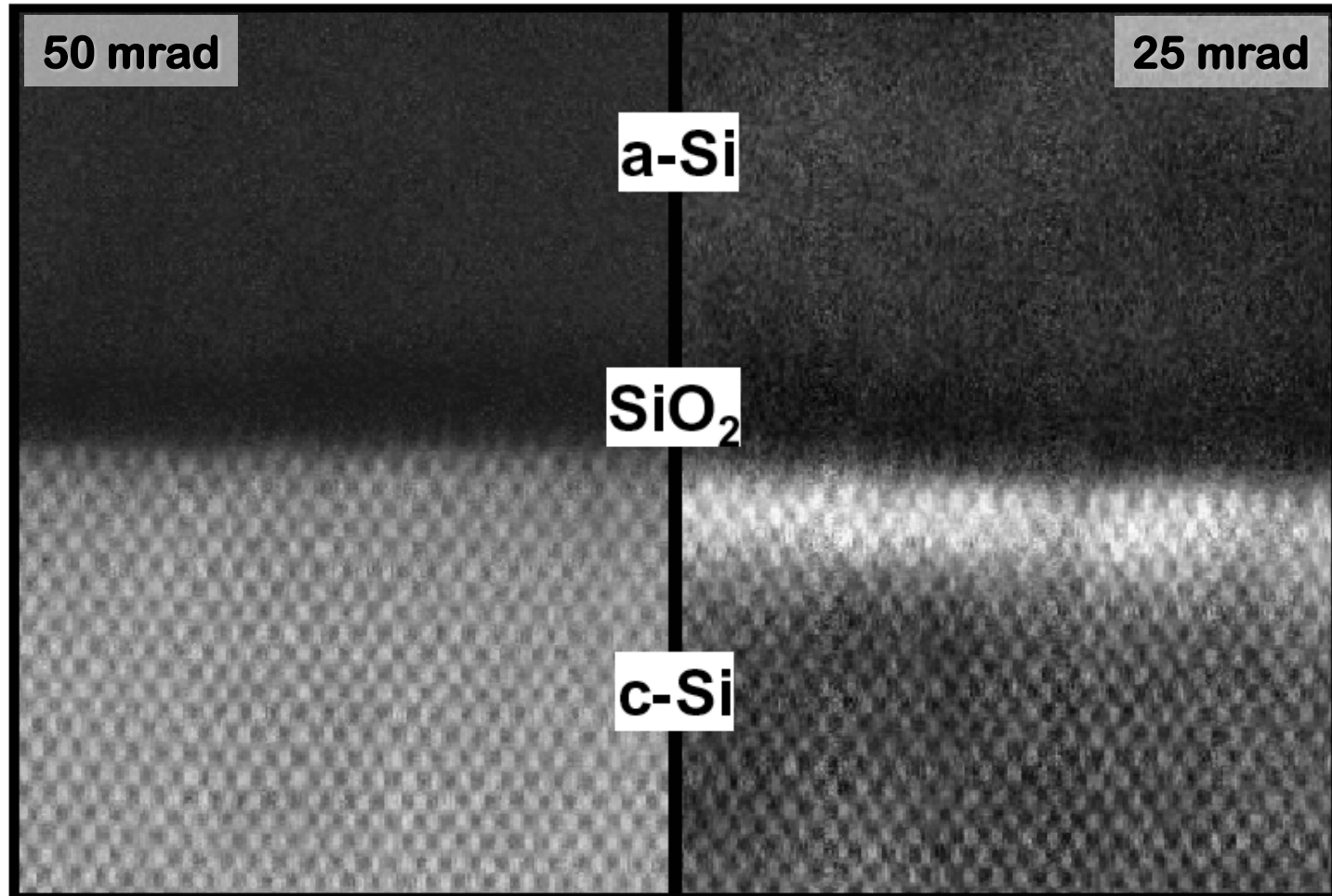


# BF-STEM



**“Reciprocity” with HRTEM images**

# Low-angle annular dark field (LAADF)

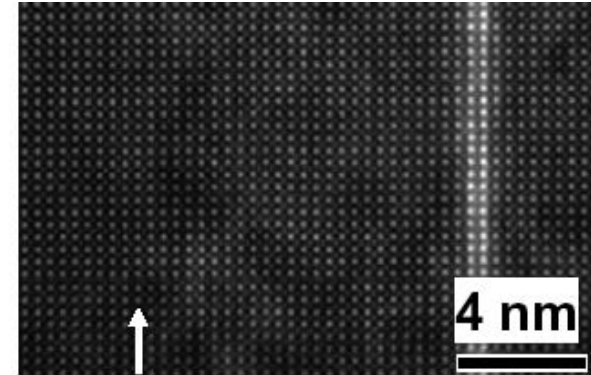


**Strain fields cause de-channeling and scattering to small angles**

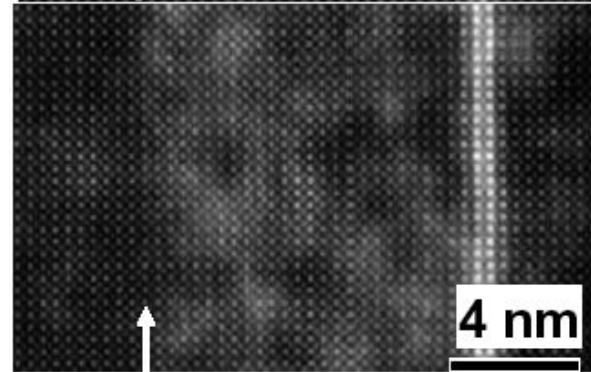
# Low-angle annular dark field (LAADF)

Here contrast is correlated with oxygen vacancies

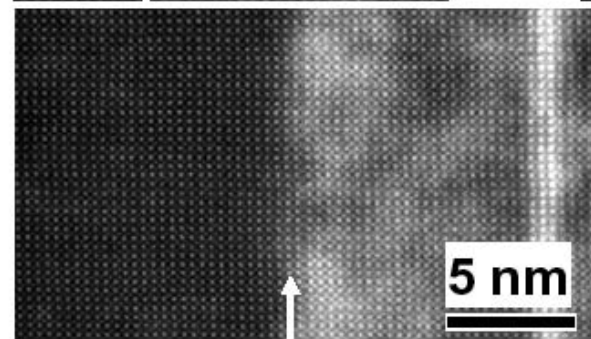
HAADF  
“Z” map



LAADF  
“Strain” map  
Thin x/s



LAADF  
“Strain” map  
Thick x/s



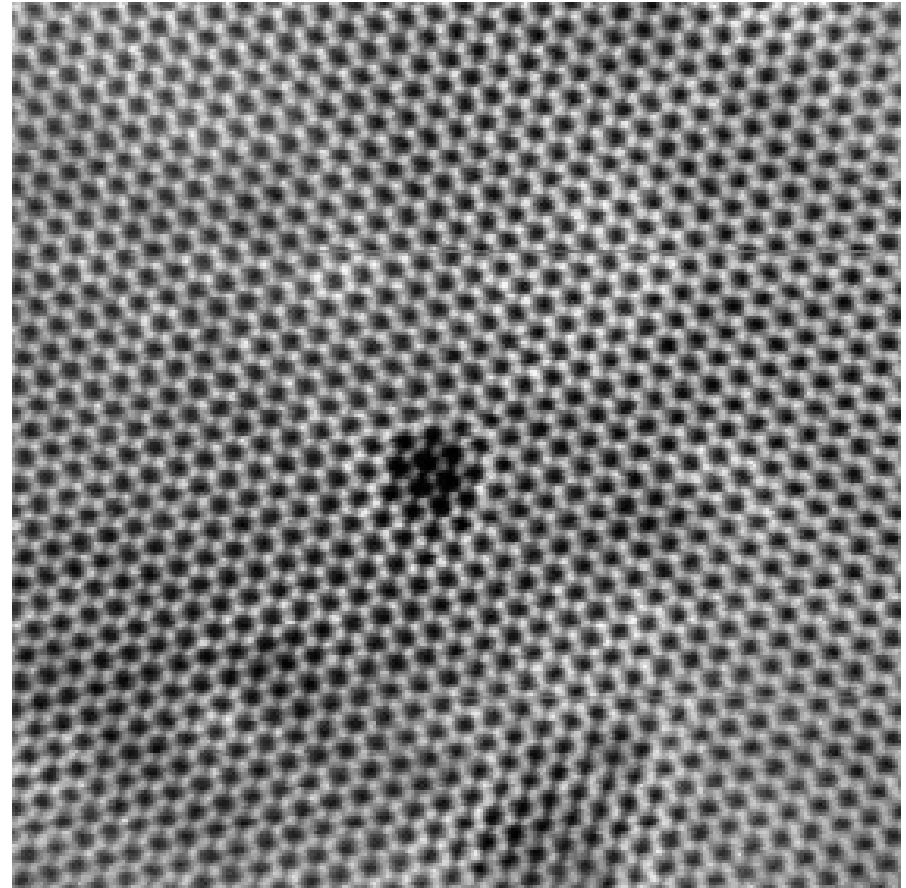
Images courtesy Dave Muller

# High angle annular dark field (HAADF)

**No contrast reversals  
with thickness**

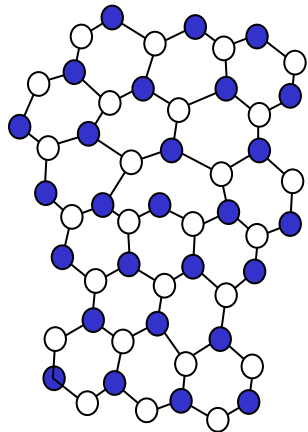
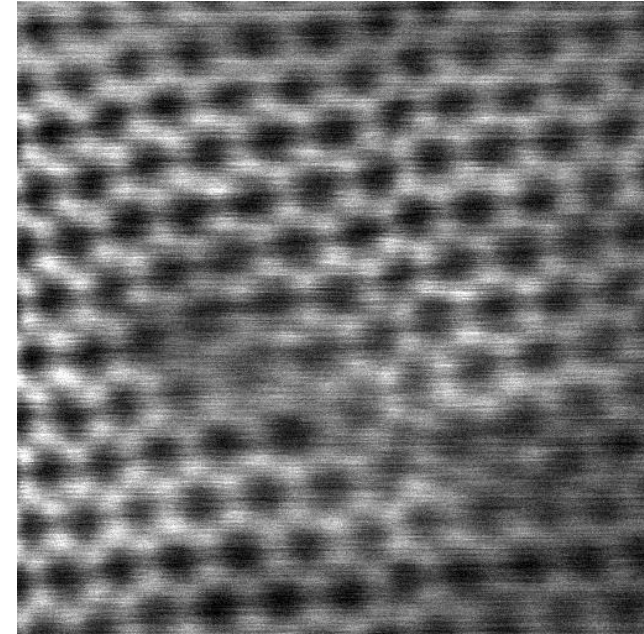
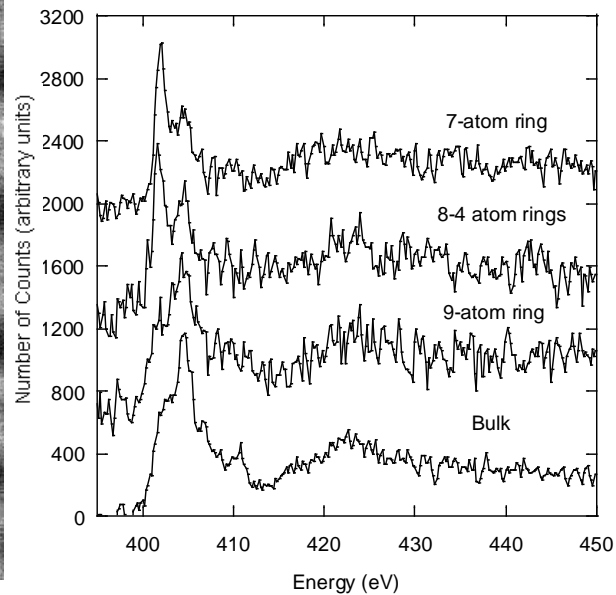
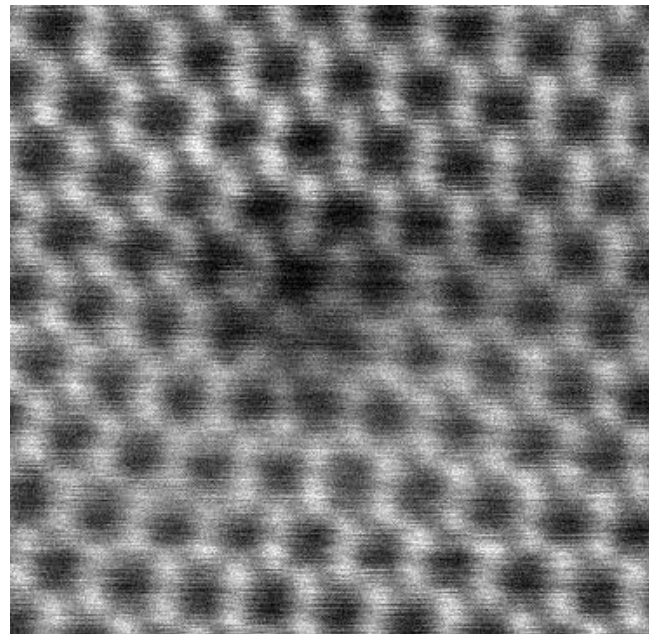
**Directly 'interpretable'  
images**

- If you see a white blob, there's an atom column there
- Caveat: the person taking (& processing) the image knew what they were doing ...



**Screw dislocation core in GaN**

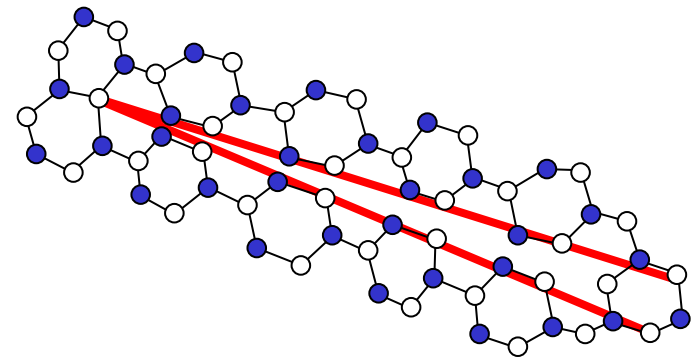
# HAADF of dislocation cores



$$\frac{1}{3} [113] \rightarrow \frac{1}{3} [110] + [001]$$

or

$$\frac{1}{3} [11\bar{2}3] \rightarrow \frac{1}{3} [11\bar{2}0] + [0001]$$



# Summary - STEM

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**Scatter to high angle is depended on atomic weight**

**Scanning a focused probe, combined with capturing this intensity can lead to an image that carries sensitivity to atomic weight differences**

**Can form atomic resolution images**

- Directly interpretable**
- Show atomically sensitive contrast**

**Can be combined with EELS to give spectroscopic information atomic column-by-atomic column**